

# Metallurgical reliability: Ensuring integrity of steels for hydrogen pipelines

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# Objective of the work

Continuous energy sector developments ➡ steels with superior properties and performance.

To fulfill these requirements ➡ understanding of the composition-processing -microstructure-property relationship, is needed.

**Objective:** to define the best practice for the production of steel for application in hydrogen high content pipelines.

# Background: different forms of cracking...

**HIC:** hydrogen induced cracking

**SSC:** sulphide stress cracking

**SOHIC:** strain oriented hydrogen induced cracking

Cracking resistance depends on both on reduction of the **initiation** sites and on reduction de factors that contribute to the **propagation** of hydrogen-induced cracking.

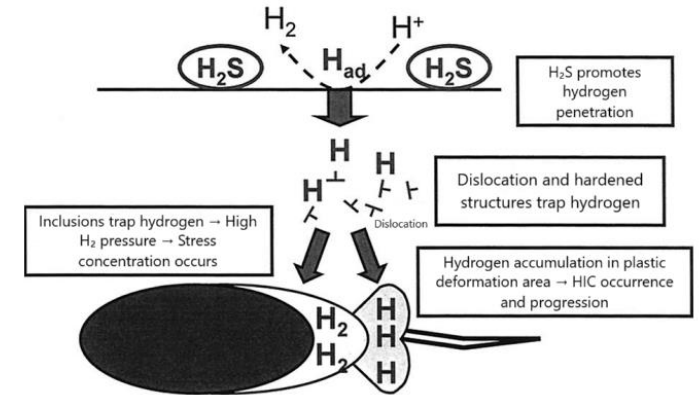


Figure 4. Material factors affecting hydrogen-induced cracking (HIC)

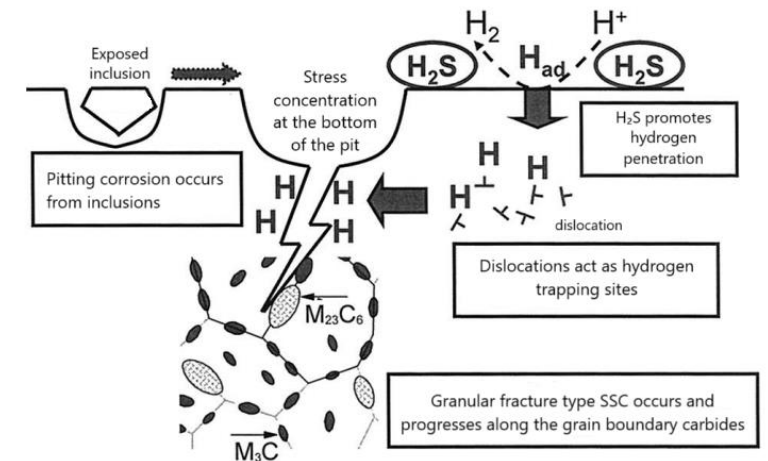
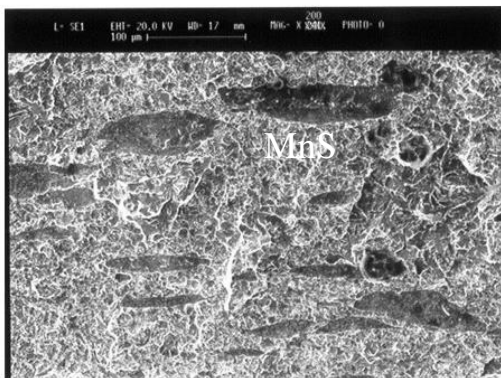


Figure 3. Material factors affecting sulfide stress cracking (SSC)

Source: Adaptation from Hydrogen Embrittlement of OCTG and Linepipes -Tomohiko Omura and Kenji Kobayashi (2011)

# Metallurgical factors for HIC resistance

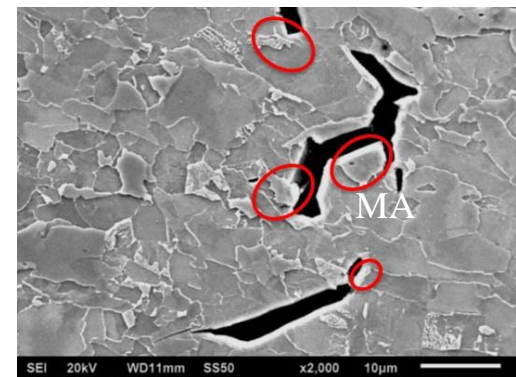
KILLING DEFECTS	IMPACT AND RISKS	TARGETS REGARDING MICROSTRUCTURE
<b>NMI - Oversize</b>	Hydrogen trap sites Stress raisers	Control severity of MnS, Al <sub>2</sub> O <sub>3</sub> , calcium oxysulfide CaAl(OS) Inclusion modification and shape control. Oversize control
<b>Coarse precipitates</b>	Initiation sites and propagation of hydrogen-induced cracking	Nano-precipitates of TiN, TiNb(C,N)
<b>Banded microstructure</b>	Nucleation and propagation of step cracks Welded defects (hook cracks)	Mainly fine FP/ NPF
<b>MA particles</b>	Crack propagation along interphase MA / matrix Impact properties highly reduced	Fine/isolated particles of martensite- austenite (MA).



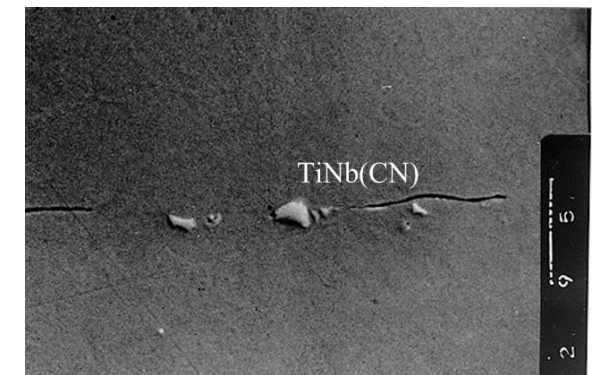
MnS on the fracture surface  
API X52



HIC step cracks

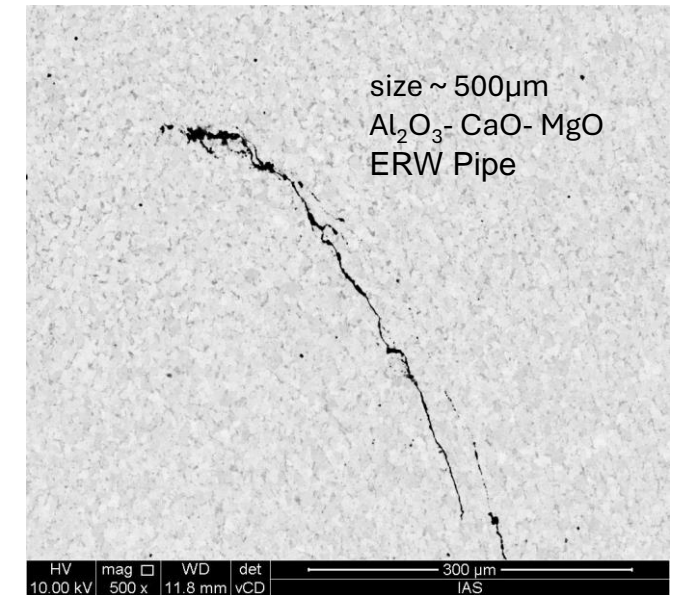
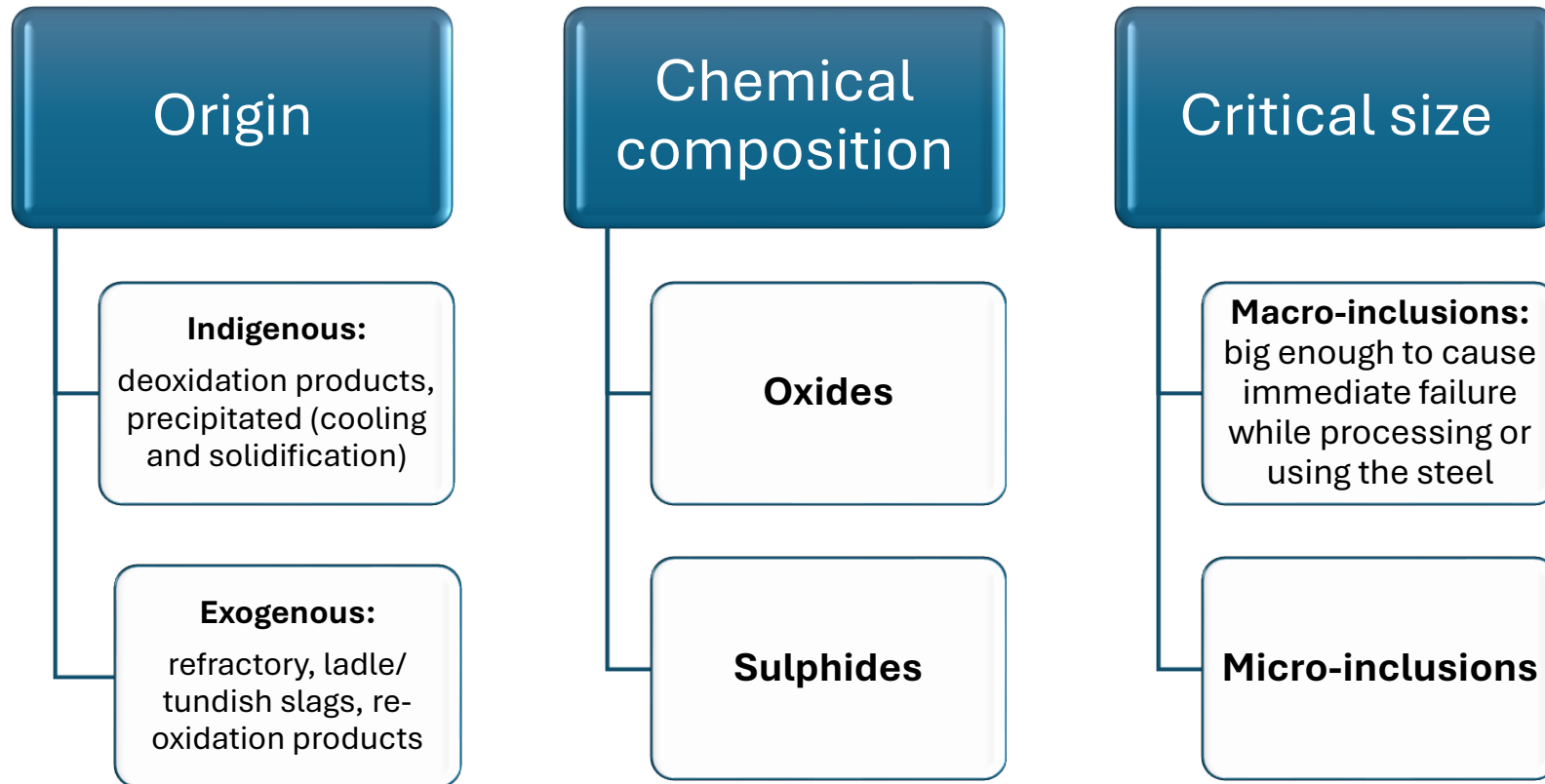


Cracks along interphase MA/ matrix  
API X70

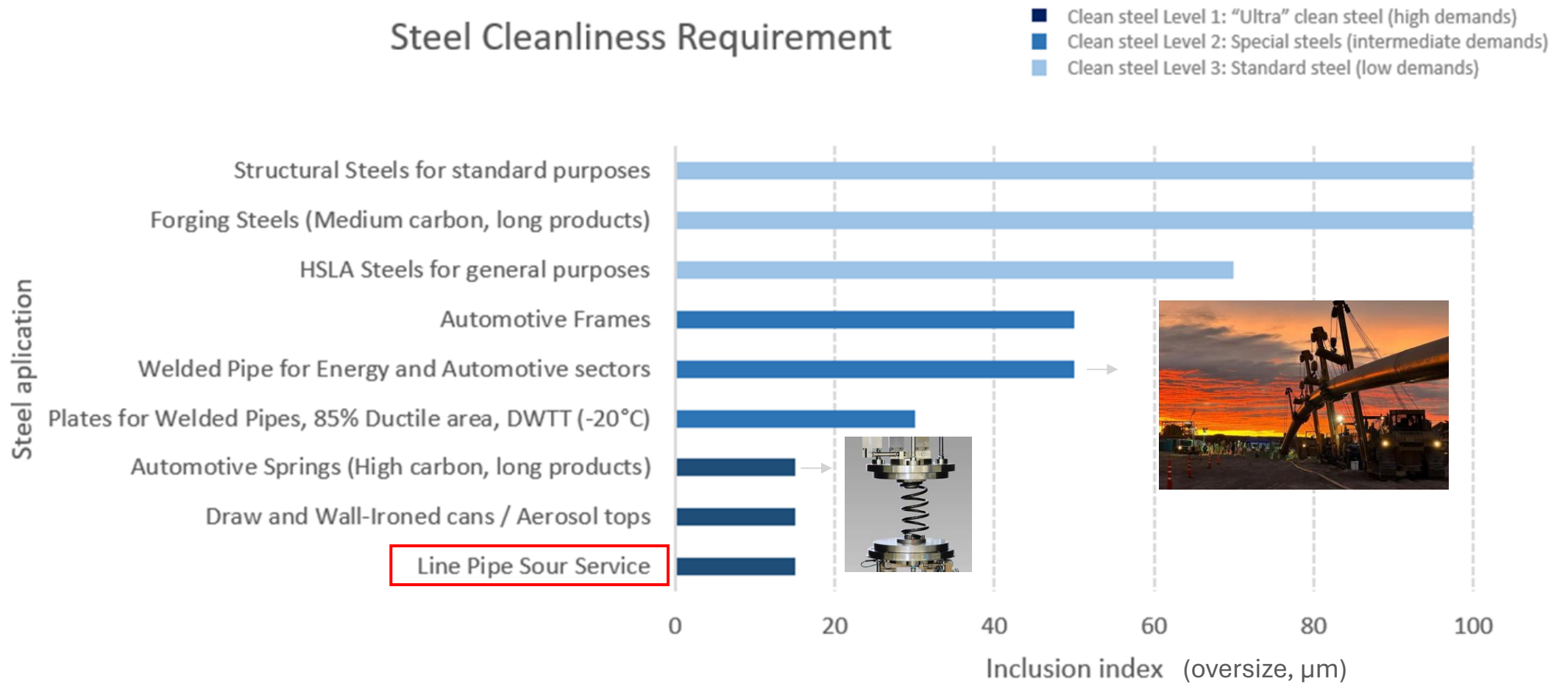


Coarse precipitates contribute to the  
propagation of HIC

# Background: the origin of NMIs

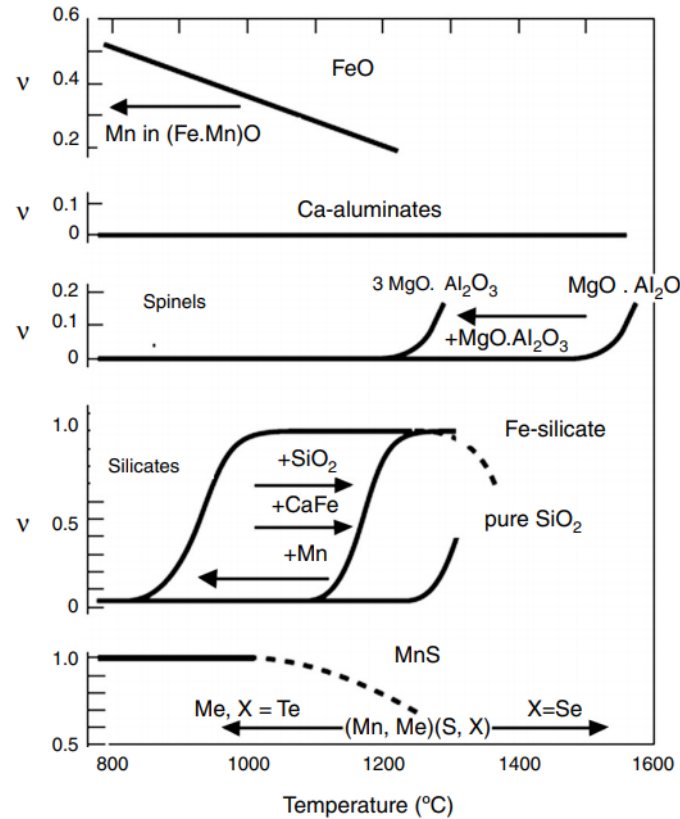


# Steel cleanliness demand by product application

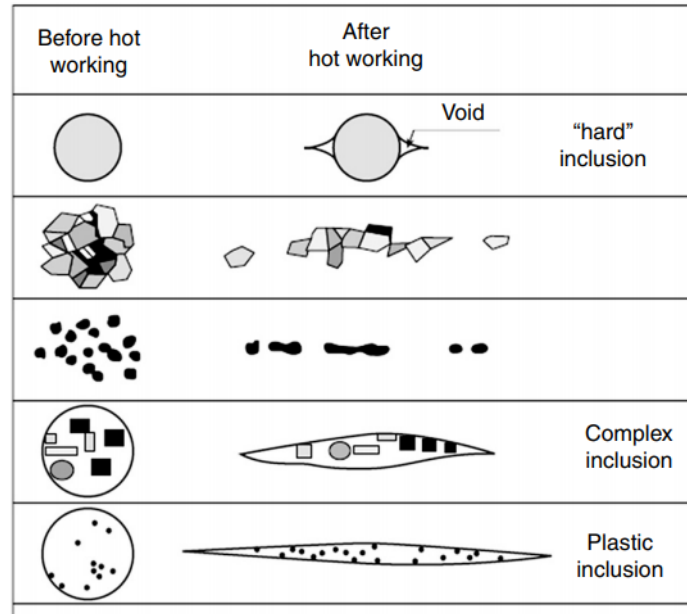


# Relevant properties of NMIs

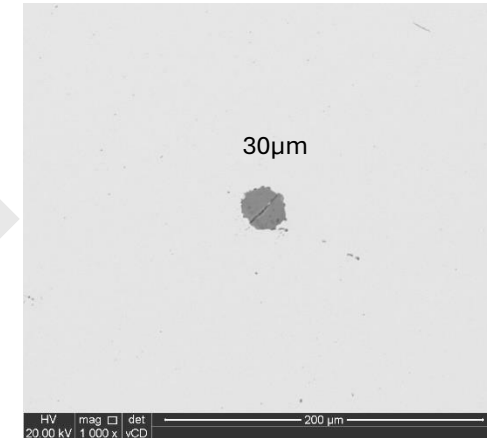
Relative plasticity  $\nu = \frac{\epsilon_1}{\epsilon_2}$      $\epsilon_1$  true elongation of the NMI, and  
 $\epsilon_2$  true elongation of the Steel matrix



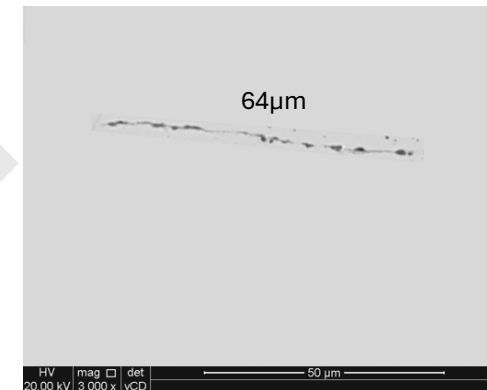
Effect of temperature on relative plasticity of NMIs, depending on their chemical composition. The curves are of semi-quantitative nature. Vasconcellos da Costa e Silva (2019)



Schematic illustration of the behaviour of NMIs after thermomechanical treatment depending on their relative plasticity at the working temperatures. Vasconcellos da Costa e Silva (2019)

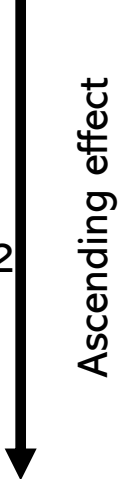


Ca-Mg aluminate:  
 60% Al<sub>2</sub>O<sub>3</sub> - 31% CaO  
 - 7% MgO - 1% MnO -  
 2% S

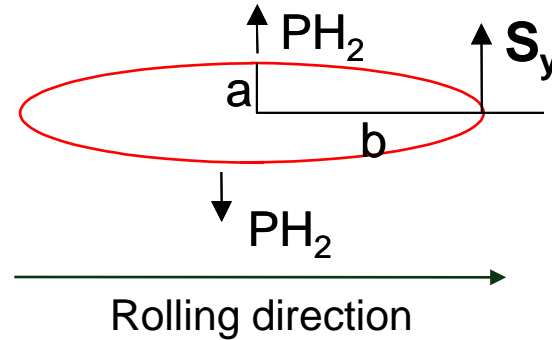


Manganese sulfide  
 70% Mn - 30% S

# Stress concentration on trap sites

Hydrogen Traps	Binding energy (eV)	
Grain boundary	0,19	
Cementite	0,19	
Intertitials	0,25	
Dislocations	0,26 - 0,32	
MnS	0,75	
Al <sub>2</sub> O <sub>3</sub>	0,82	
TiC	0,90	

## 1 Elliptical section:



$S_y$  = Stress concentration on the tip

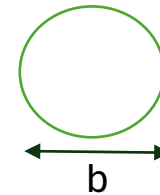
$PH_2$  = internal pressure of H<sub>2</sub>

When  $a/b = 0,1$

$$S_y/PH_2 = \mathbf{16}$$

$$S_y = \frac{PH_2}{(a/b)^{1,2}}$$

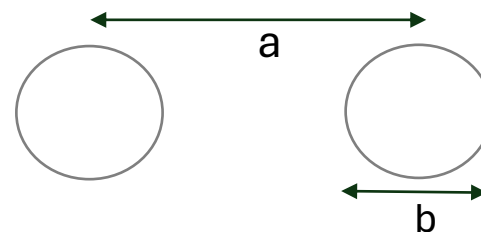
## 2 Round section:



In this case  $a = b$

$$S_y/PH_2 = \mathbf{1}$$

## 3 Influence of inter-particle distance:



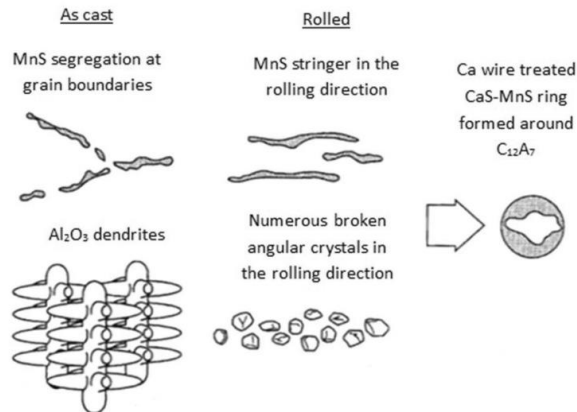
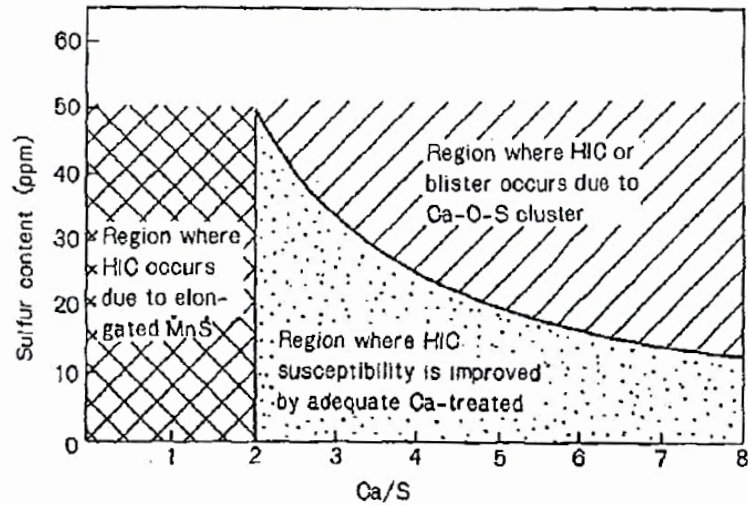
Stress concentration factor= 1,85

$$a = 2.b$$



# Inclusion modification and shape control

Influence of Ca/S ratio and sulfur content on improving the HIC resistance of a calcium-treated steel tested in solution of a0,5% acetic acid – 5% NaCl and H<sub>2</sub>S  
 A. Ikeda *et al.* Sumitomo Search, N°26 (1981)



Schematic illustration of inclusion morphology with calcium- treated steel. Turkdogan, Fundamentals of Steelmaking, The Institute of Materials (1996).

## HIC resistance:

Morfology and shape control of Al<sub>2</sub>O<sub>3</sub> and MnS, Ca/S > 2

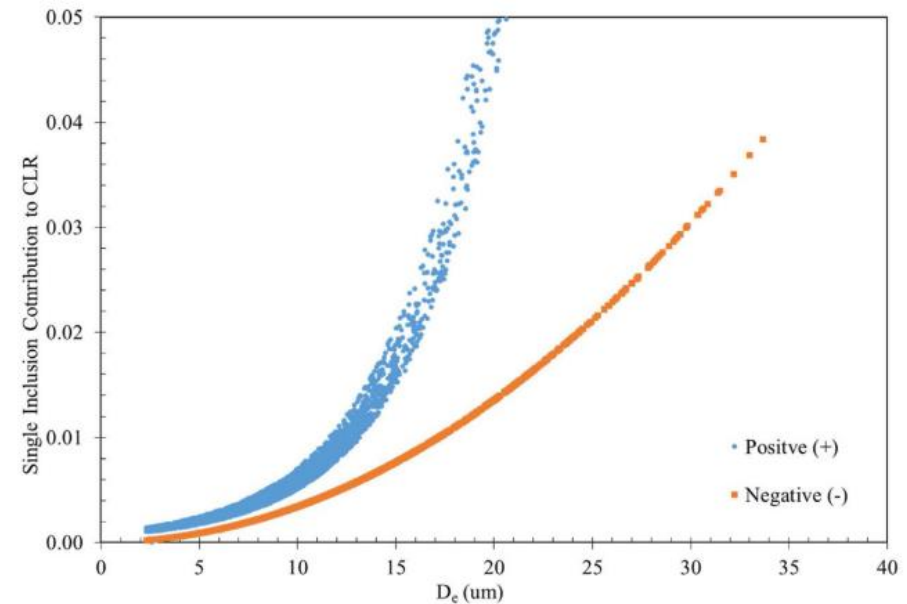
$$\text{Globulization index (ACR)} = \text{Ca}_{\text{eff.}} / \text{Ca}_{\text{combined with S}} = \text{Ca}_{\text{eff.}} * 1,25.S$$

$$\text{Ca}_{\text{effective}} = \text{Ca}_{\text{total}} - \text{Ca}_{\text{calcium oxide}}$$

S < 0,002% (aim < 10ppm)

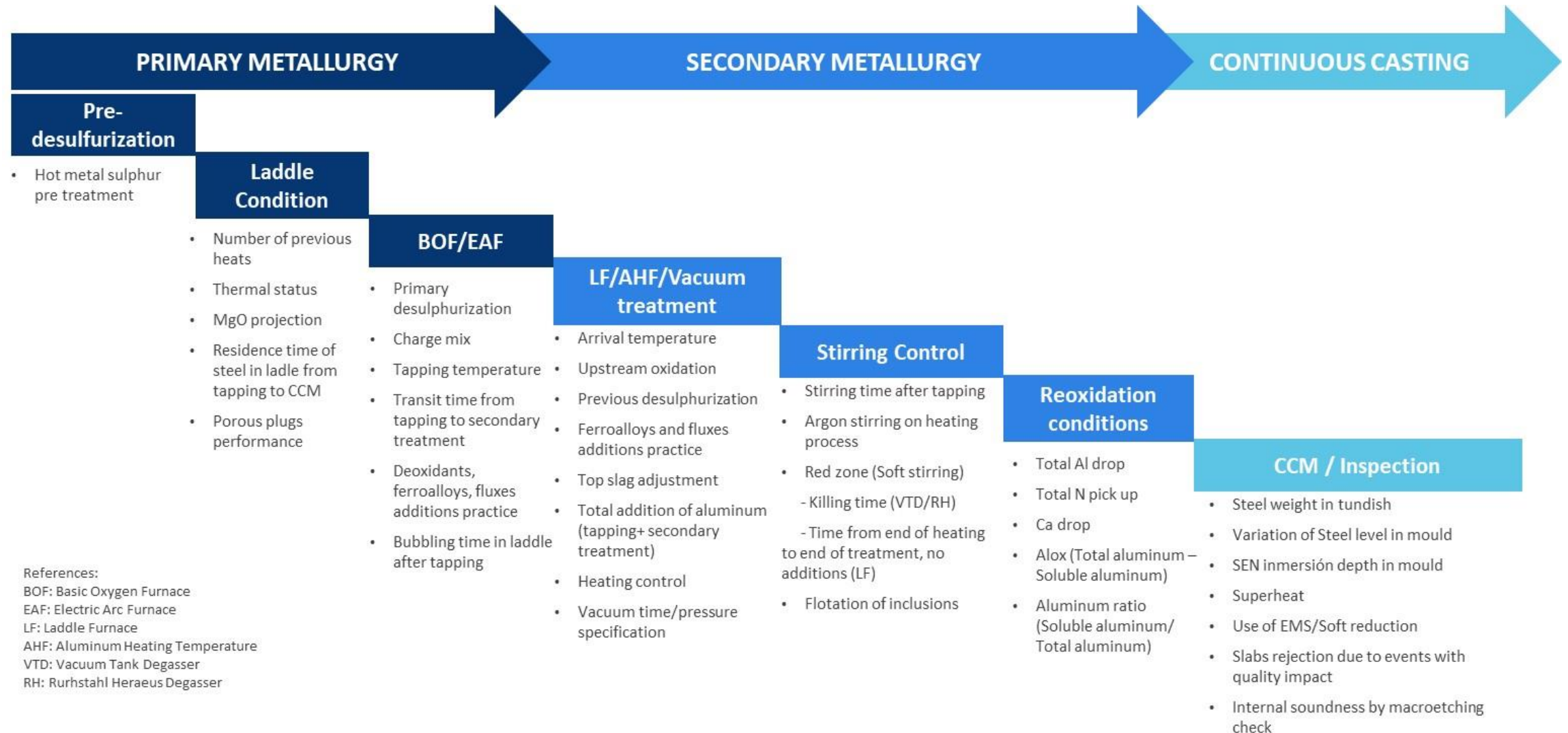
O < 30 ppm (aim < 20 ppm)

OS (De) < 15µm



Effect of Oxide De on increasing CLR and decreasing CLR (trapping term) for a single inclusion. Ouhiba *et al* (2023)

# Identification of key process parameters

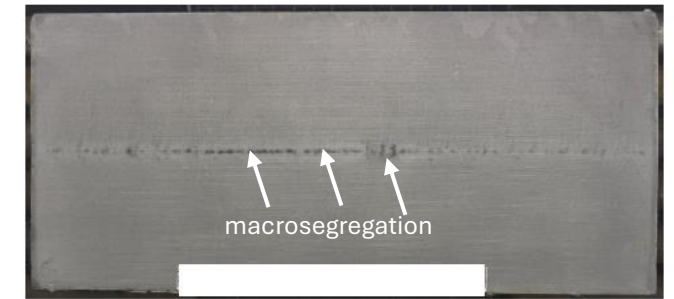
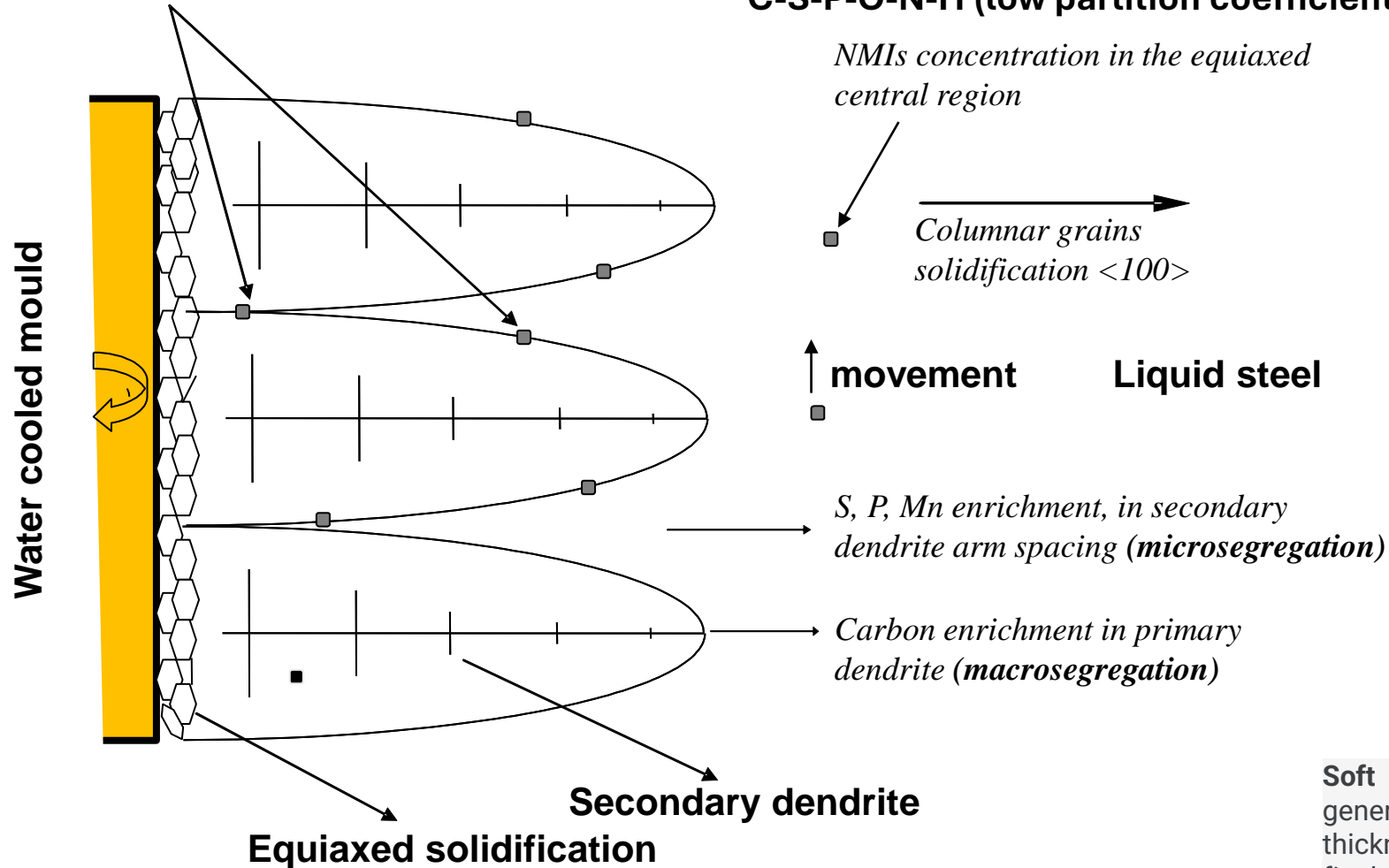


# Schematic illustration of solidification<sup>1</sup>

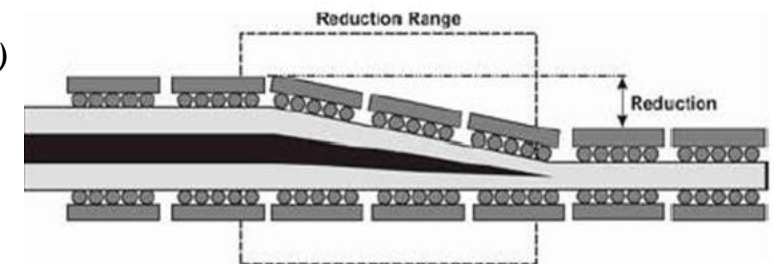
Interdendritic NIMs (and TiN)

Elements conducive to microsegregation:

**C-S-P-O-N-H (low partition coefficient values)**



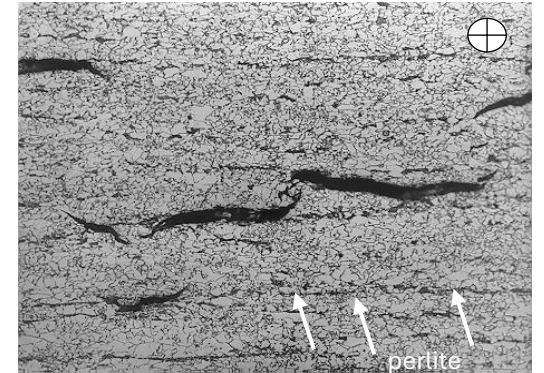
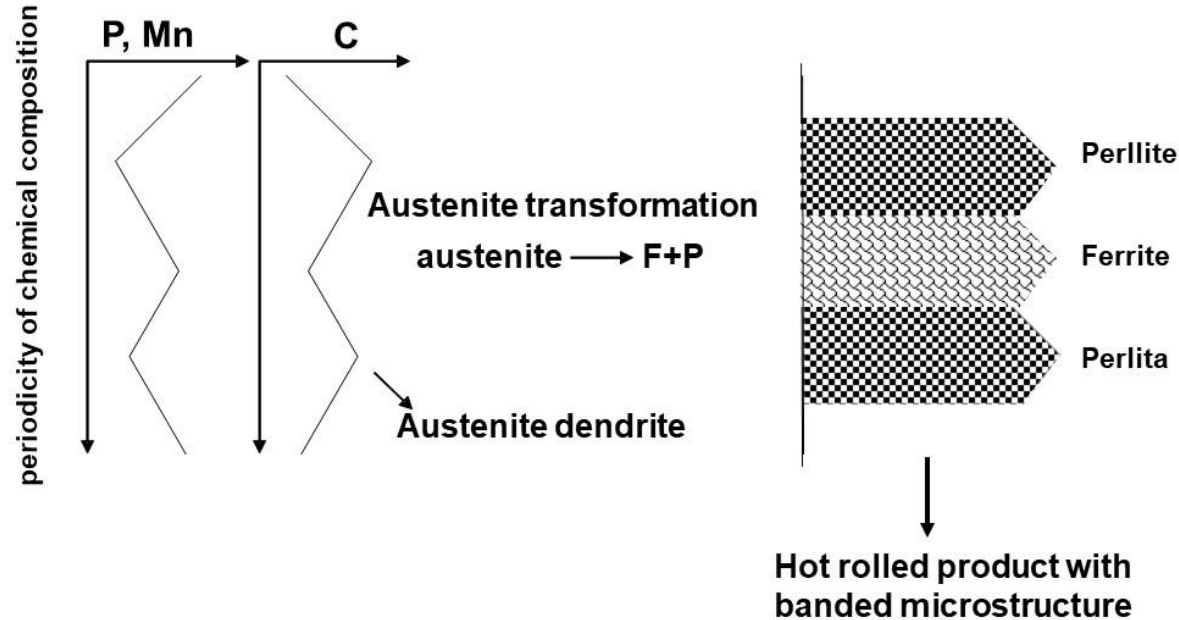
Macroetching – 20% ammonium persulfate



**Soft reduction (SR):** established technology and is generally understood as a reduction (~2% of the thickness) of the cross-section of the strand (near to the final solidification point)

<sup>1</sup> Adapted from R. Ratnapulli, ABM (2007)

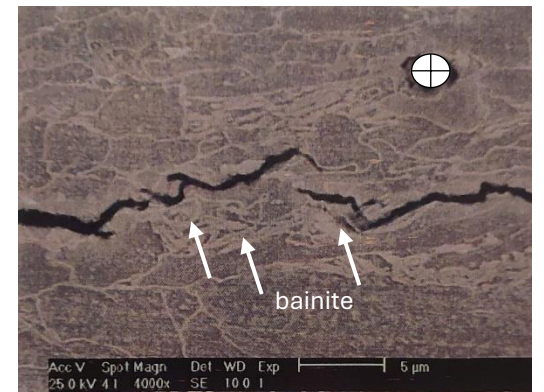
# Microsegregation and banding



HIC step cracks

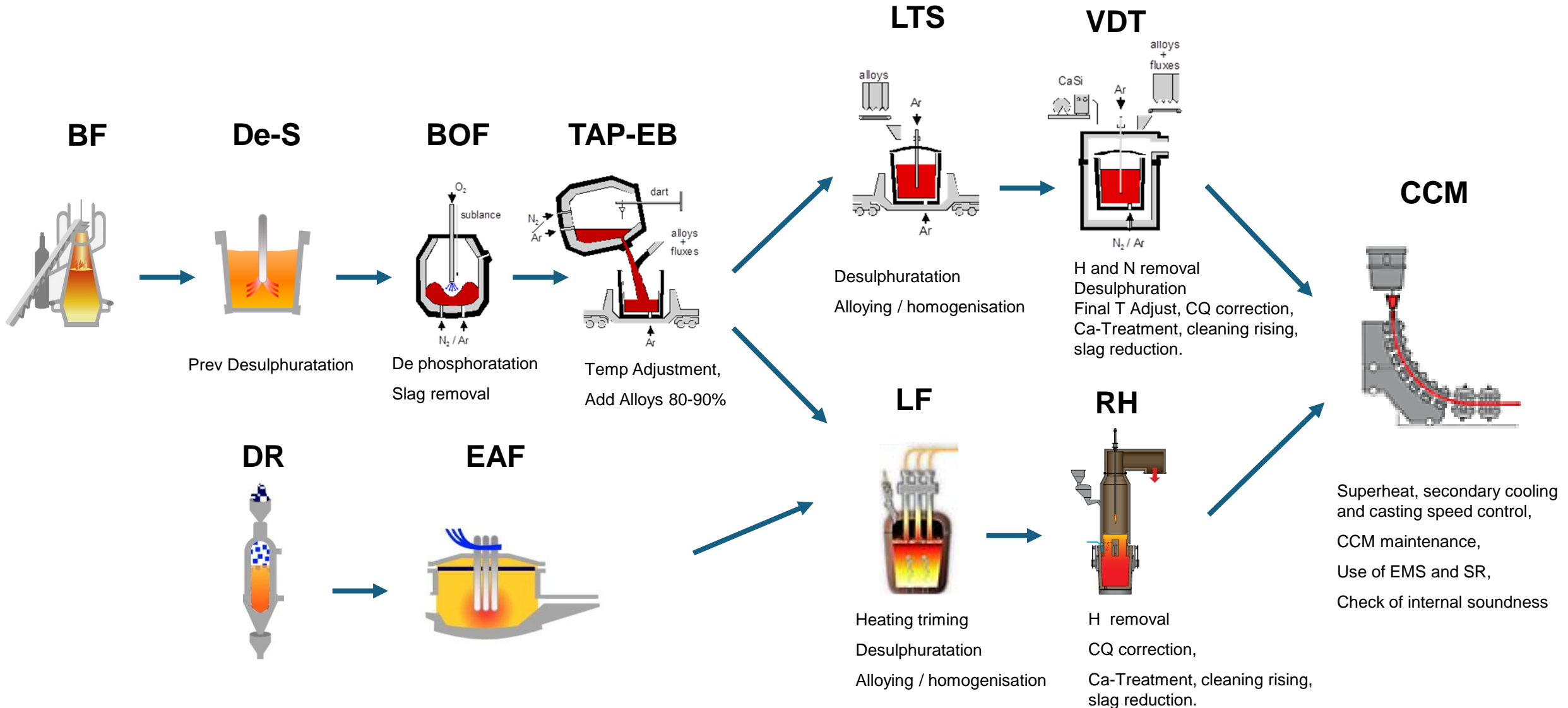
## Minimization of banding severity

- Reduction of C, Mn and P contents
- Control of continuous casting parameters (superheat, secondary cooling, use of EMS and SR)
- Redistribution of C with accelerated cooling after hot rolling

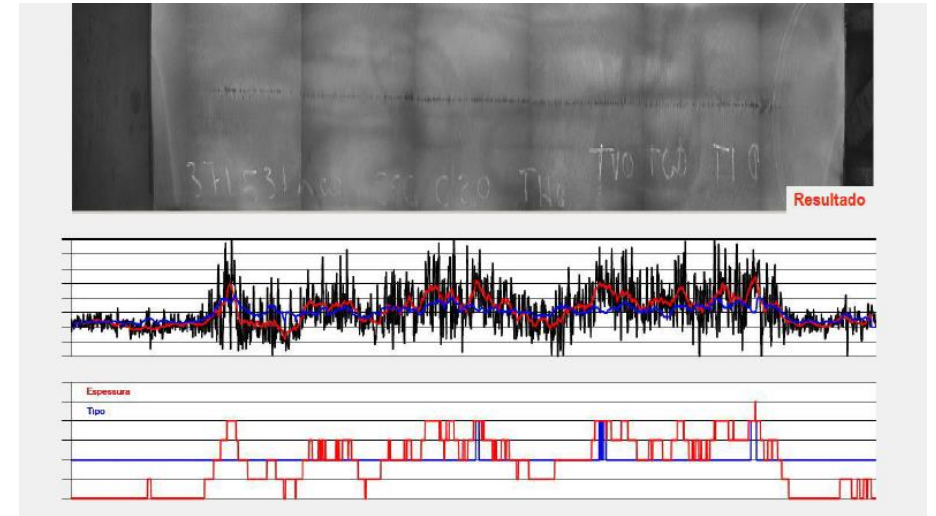
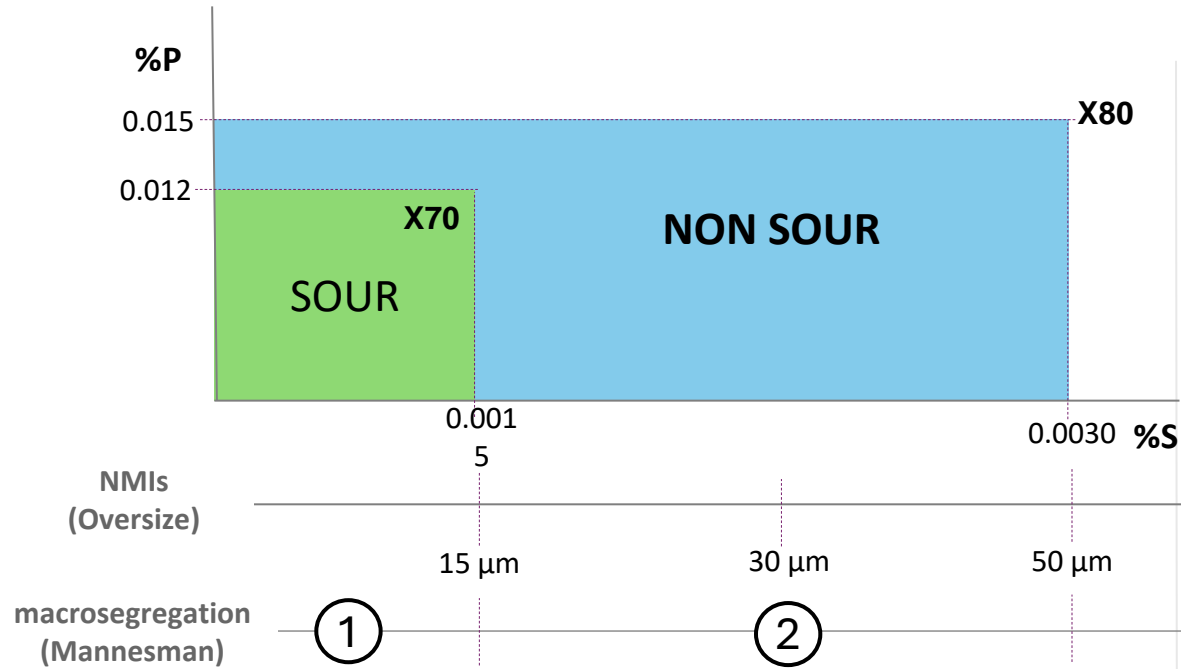


Crack propagation in mixed softer ferrite (180HV) and bainite (> 210HV)  
API X60, NbTiCuNi

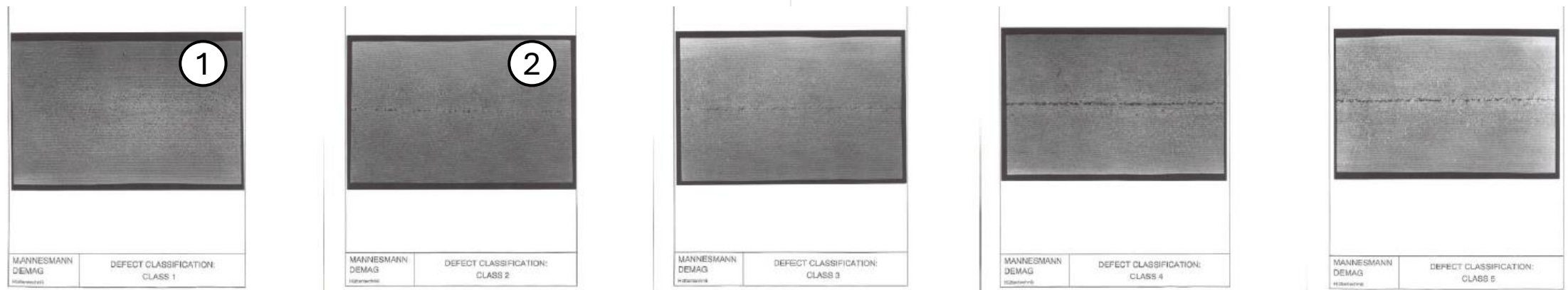
# Technical approach : steelmaking overview



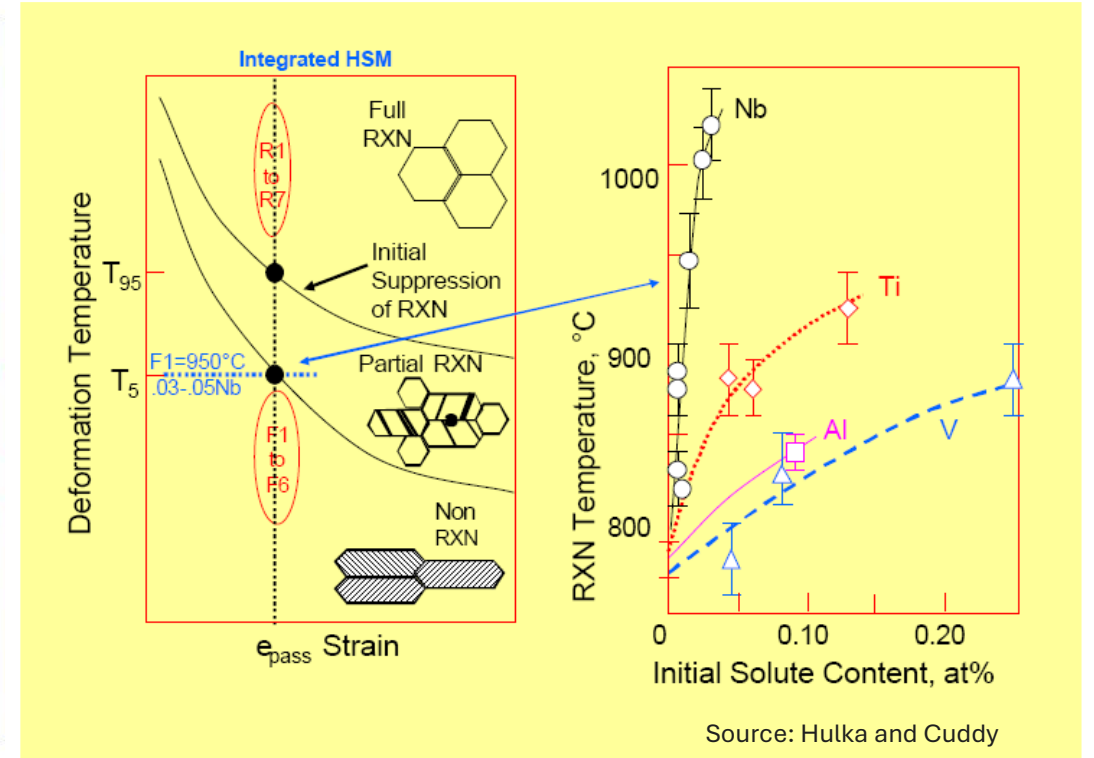
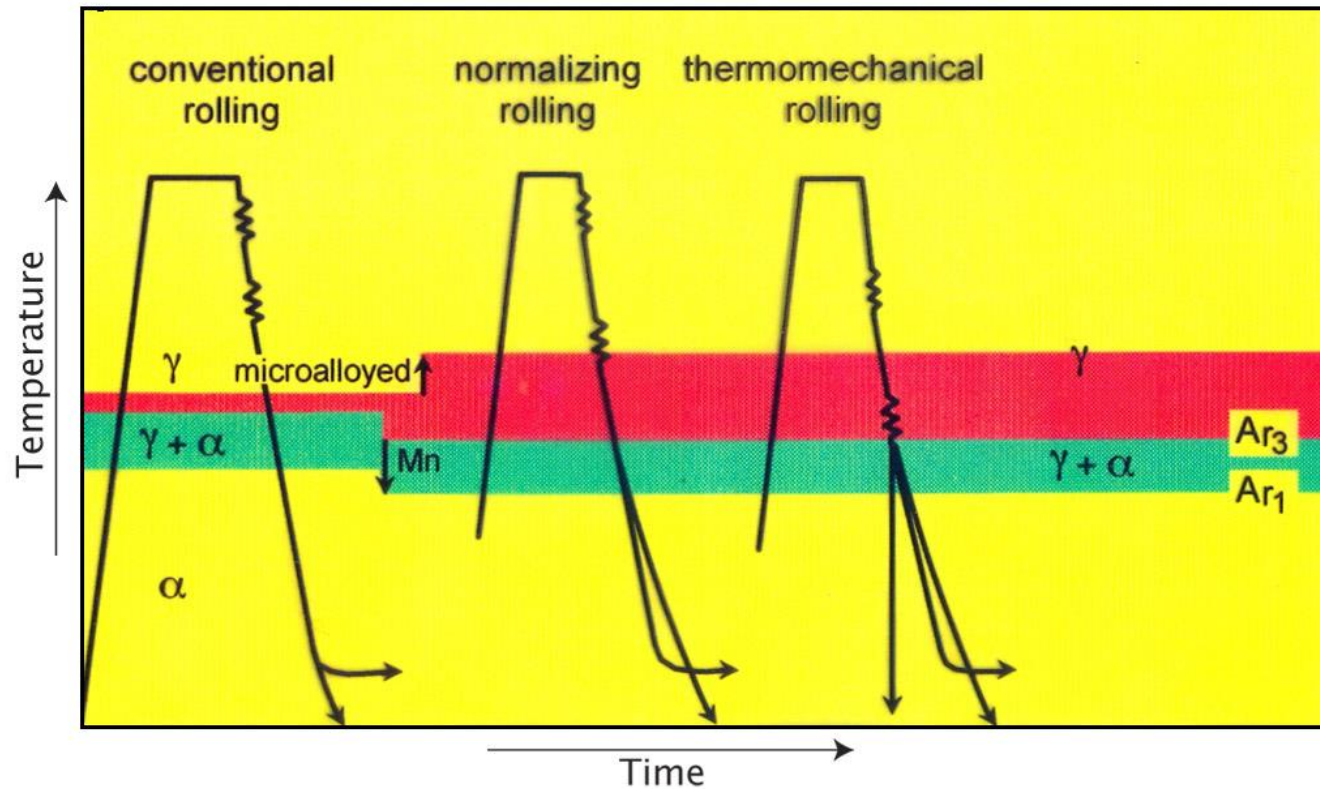
# Residuals, steel cleanliness and soundness



Source: F. Ramstorfer et al. - COMPUTATIONAL CLASSIFICATION OF CENTRAL SEGREGATION OF PLATES BY IMAGE PROCESSING OF BAUMANN PRINTS, ABM Proceedings (2014)



# Thermomechanical processing (TMP)

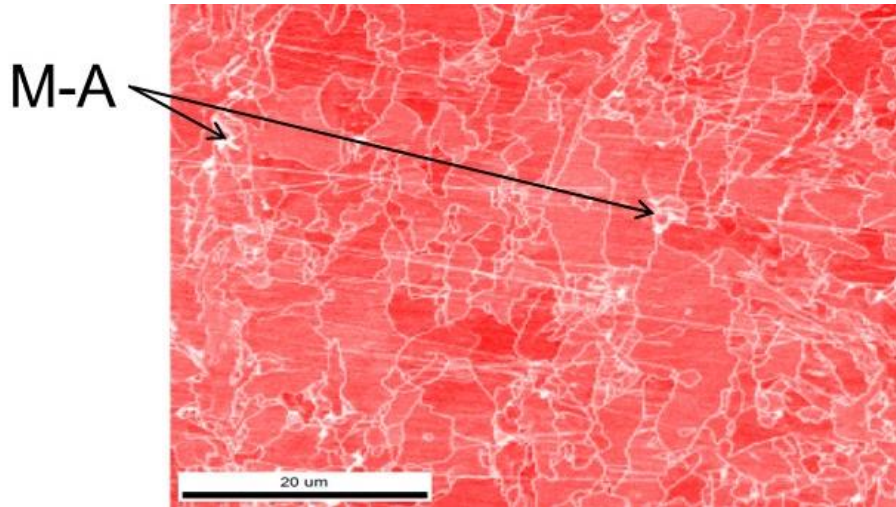


**Goal of the TMP:** fine grain, isolated MA constituent

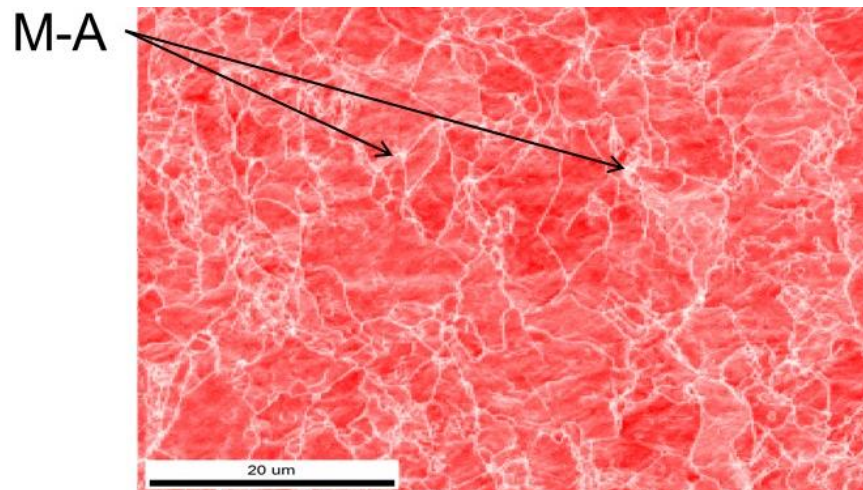
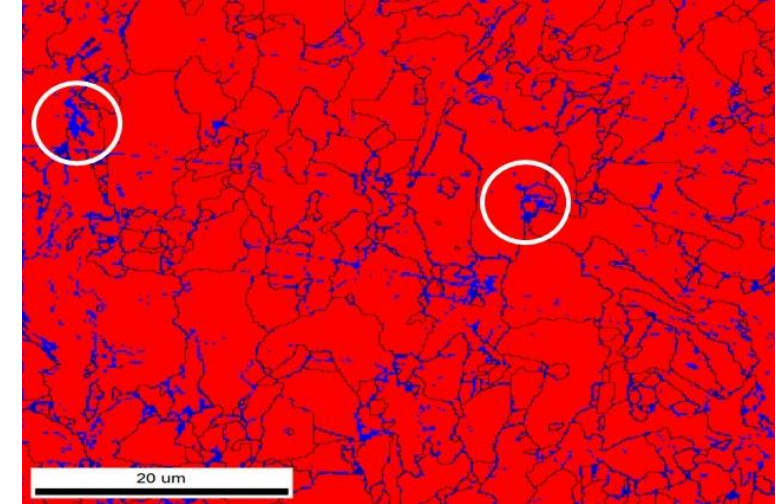
- Proper alloy design to increase the processing window as defined by  $T_{nr}$  and  $T_{ar3}$ ,
- Processes conditions control during hot rolling,
- Transformation control

Source: Hulka and Cuddy

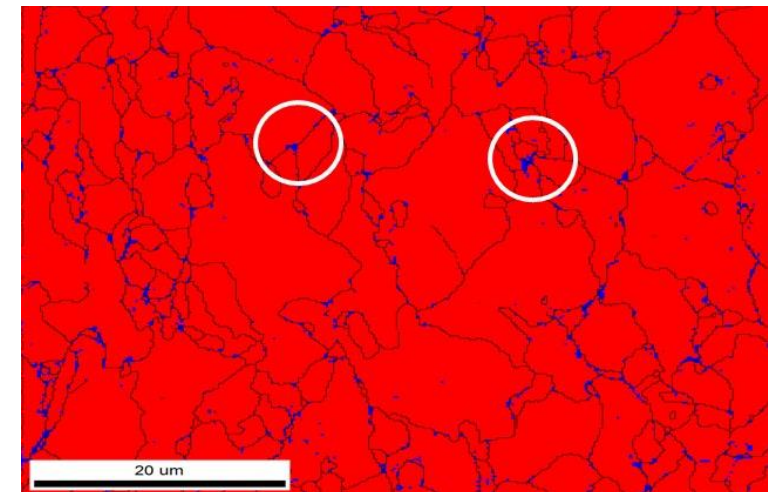
# MA control by TMCP, API X70 (t=12.7mm)



5.7% MA  
DA < 85%  
-20° C



1.2% M-A  
DA = 100%  
-40° C

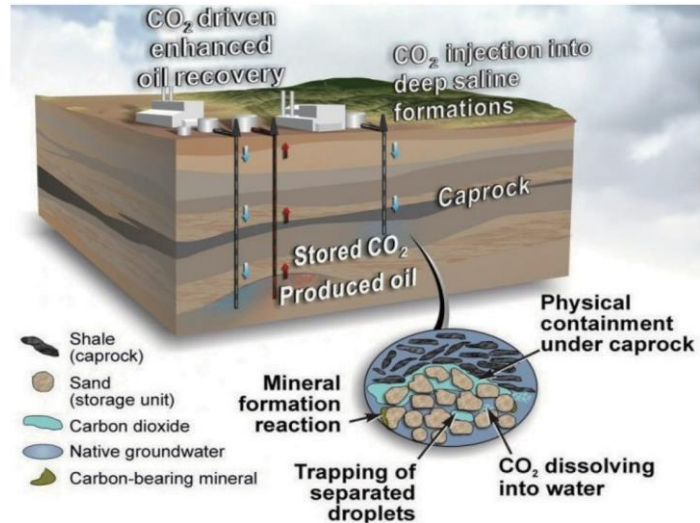




# Metallurgical factors to be controlled: resume

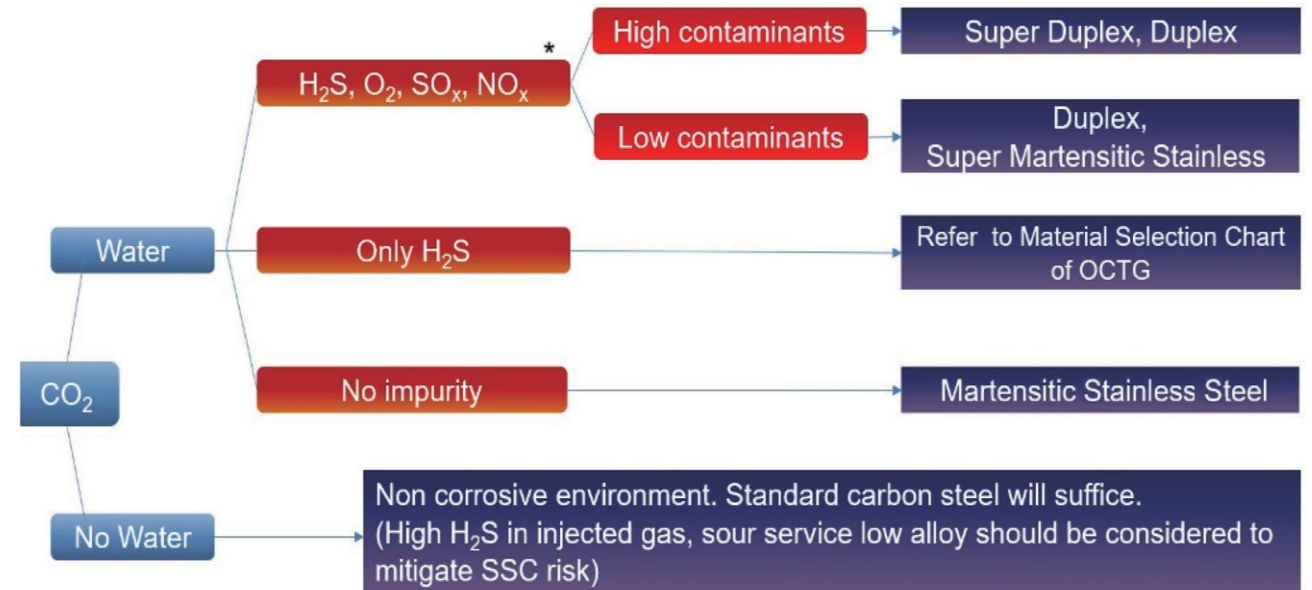
KILLING DEFECTS	TARGETS REGARDING MICROSTRUCTURE	SOLUTIONS & CHALLENGES
<b>NMI - Oversize</b>	Control severity of MnS, Al <sub>2</sub> O <sub>3</sub> , calcium oxysulfide CaAl(OS) Inclusion modification and shape, maximum oversize (OS) control	<b>Best practice for production of ultra-clean steels</b> (low S, H contents, Ca-treated steels) OS < 15µm, globular oxide type (avoid sulfides and alumina types).
<b>Coarse precipitates</b>	Nano-precipitates of TiN, TiNb(C,N)	Low Nitrogen steels, Ti/N= 2.5-3.0 (sub-stoichiometric ratio), mean particle size: 100nm Use of Electromagnetic stirring (EMS) and Sof reduction (SR) at the CCM
<b>Banded microstructure</b>	Mainly fine FP/ NPF	<b>Steel design</b> : reduction of P, Mn and C contents.  <b>Best practice for the casting high quality of slabs:</b> - Superheat, secondary cooling and casting speed control, - Position control and alignment of the CCM equipment, - Use of EMS and SR reduction in strand to reduce micro & macrosegregation, - Check of internal soundness (Mannesmann 1 maximum severity).
<b>MA particles</b>	Fine/isolated particles of martensite- austenite (MA).	<b>Best practice for thermomechanical rolling of microalloyed steels:</b> proper alloy design, processes conditions during hot rolling and transformation control - Select alloy elements (Nb, Ni, ...) → Reduce austenite/ ferrite starting temp. (Tar3) - Increase reduction rate in last passes → Reduce final effective austenitic GS, big ΔT (Tnr-Tar3) - Reduce finishing temperature. Accelerated cooling → redistribution of carbon during transformation - Hardness < 210 HV (hard spot control)

# Corrosion in CO<sub>2</sub> environment (CCS)



Schematic image of CCS process

Component	Corrosion effects
CO <sub>2</sub>	• Balance Gas pH drop out
O <sub>2</sub>	• Oxidant: risk of localized corrosion • Elemental Sulfur (S <sub>0</sub> ) produced by reaction with H <sub>2</sub> S. → risk of localized corrosion
SO <sub>2</sub>	• Sulfuric acid (H <sub>2</sub> SO <sub>4</sub> ) produced. → pH <u>drop</u> out
NO <sub>2</sub>	• Possibility of producing Nitric acid (HNO <sub>3</sub> ) → pH <u>drop</u> out • Oxidant: risk of localized corrosion
H <sub>2</sub> S	• Risk of Hydrogen Embrittlement because H <sub>2</sub> S acts as a poison in Hydrogen penetration



# Conclusions:

- Research today focuses on overcoming technical concerns related to pipeline transmission, including the potential for hydrogen to embrittle the steel and welds used to fabricate the pipeline
- To fulfill these requirements a **proper balance** of toughness, mechanical strength, weldability and resistance to cracking of the pipe in the field is needed.
- Cracking resistance depends on both on reduction of the **initiation** sites and on reduction of factors that contribute to the **propagation** of hydrogen-induced cracking.
- By adopting and implementing a **tailor-made Best Practice** at the steelmaking and hot rolling processes steel suppliers can produce significantly cleaner steels with the necessary microstructure
  - Due attention must be given for improving **steel cleanliness** and reducing the **macro and microsegregation** of elements such as S, P, O, N and H, in order to improve the behavior of the base metal and the welded joint.
  - **Advanced process control** of key process variables by using IA, ML, expert-systems are highly recommended for quality prediction of the steel supplied.
- This leads to improved hydrogen pipelines with a novel approach based on **metallurgical reliability**.